Spin state, charge carrier and thermal transport in cobalt perovskites

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Abstract In continuation with both the thermoelectric evaluation and exciting physics of cobalt based perovskites we studied the interplay of thermal and electric transport, the spin state and the local symmetry of the coordination of Co ions. The comprehensive experimental examination of systems covering a wide range of $\text{Co}^{2+}/\text{Co}^{3+}/\text{Co}^{4+}$ ratio involving the cubic *Pm*3*m* and rhomohedral *R*-3*c* phases, with straighten Co-O-Co bond, orthorhombically distorted *Pbnm* perovskites and 2D perovskites *I*4/*mmm* of La_{2-x}Sr_xCoO₄ type, was performed in a wide temperature domain up to 1000 K. The observed results, referring namely the thermoelectric power and thermal conductivity, are discussed with respect to possible spin states and orbital degeneracy of Co²⁺/Co³⁺/Co⁴⁺ species.

1. Introduction

Cobalt perovskites have been recently revised with respect to their promising thermoelectric potential- see.g.[1]. Furthermore, in spite of already 50 years research of the spin state of cobalt ions and the nature of the broad magnetic-electronic transitions in Ln1-xAxCoO3 perovskites, the unequivocal physical background is still lacking [2]. In the ground state of prototype LaCoO₃ there is no magnetic moment on Co ions, which implies that all trivalent cobalt are in the diamagnetic, low-spin state (LS, $t_{2g}^{6}e_{g}^{0}$, S = 0)-see Fig.1. With increasing temperature the system becomes paramagnetic, the susceptibility increases rapidly and shows a broad maximum around 100 K. Another pronounced anomaly in the susceptibility accompanied with a significant decrease of the resistivity is observed at 500 K. The 100 K transition was first interpreted by Goodenough as a change of 50% of cobalt ions from the low-spin ground state to high-spin state (HS, $t_{2g}^4 e_g^2$, S = 2) -see Table 1. Simultaneously, the 500 K transition was assigned to an order-disorder transition, where the eg electrons become itinerant and destroy the LS-HS superstructure. The latter was, however, never observed experimentally when a dynamic resonance of LS and HS Co3+ states was proposed subsequently [3]. More recent experiments of photoelectron emission and X-ray absorption spectroscopy, supported by band structure calculations led to an alternative interpretation of the 100 K transition as a stabilisation of an intermediate spin state (IS, $t_{2g}{}^{5}e_{g}{}^{1}$, S = 1). This is an orbitally degenerate (Jahn-Teller active) state in which both the eg electron and t2g hole may be coupled with the same distortion mode of the CoO6 octahedron of the Eg symmetry. Using the results of LDA+U, Korotin et al. [4] explained stabilization of Intermediate Spin state over High Spin state due to the large hybridization between Co-3d (eg) and O-2p orbitals. The physical reason for this transition is a competition between crystal-field energy Δ_{cf} and intraatomic (Hund) exchange energy Δ_{ex} (Fig. 1). In the ground state Δ_{cf} is only slightly larger then Δ_{ex} so the energy of the excited magnetic state of Co ion is relatively small and with increased temperature its population increases resulting in an increase of magnetic susceptibility. To demonstrate the complexity of the possible ground states (spin and orbital) the total degeneracy for Co²⁺, Co³⁺ and Co⁴⁺ species is shown in Table 1.



Fig 1. Scheme of energy levels for Low Sin state of Co³⁺ ions in LaCoO₃

When La ions are substituted in perovskite cobaltites by smaller rare earth and alkaline earths elements (inducing Co⁴⁺ formation), magnetic properties show significant changes as demonstrated in Fig.2 using our data.



Fig2. The temperature dependence of the DC magnetic susceptibility for $La_{1-x}Ba_xCoO_3$ and $Y_{1-x}Ca_xCoO_3$. In the LaCoO₃ the dominating interactions are antiferromagnetic while the $La_{0.6}Ba_{0.4}CoO_3$ is ferromagnetic due to introduction of Co⁴⁺ species.

Coherently the ⁵⁹Co Knight shift measurements showed that Co ions remain to be in the Low Spin state up

to the temperature where the gradual metal-insulator transition is observed For (e.g NdCoO₃ up to 580 K). Our X-ray diffraction study of lanthanide cobaltite perovskites $LnCoO_3$ (Ln=Nd, Gd, Dy and Y) led to similar conclusion- that "at room temperature all cobalt ions are in the Low Spin state regardless of the Ln atomic number". Moreover magnetic and electric measurements for Ho_{1-x}Ca_xCoO₃ system [5] showed that "there is no apparent transition between Low Spin and High Spin of Co³⁺ ions in 300-900 K temperature range". All these data indicate that Low Spin state of Co³⁺ ions in lanthanide cobaltite perovskites with smaller rare earth ions becomes more stable comparing with LaCoO₃ and transition temperature increases dramatically.



Table 1 Possible spin states and total degeneracy of ground-states of Co^{2+} , Co^{3+} a Co^{4+} species (S=Spin only number)

In following paper we prsent the experimental data of the thermal conductivity, thermoelectric power, electrical resistivity and magnetization of various 3D perovskites $Ln_{1-x}A_xCoO_3$ (Ln=La,Y and A=Ca, Sr and Ba) and the 2D $La_{2-x}Sr_xCoO_4$ systems for 0.5 < x < 1.5. We perform the measurements up to highest temperatures (~1000 K) and we analyse, with respect to possible spin and orbital states, the plausible mechanism of the charge carrier transport both for electron (Co²⁺) and hole (Co⁴⁺) doped Co³⁺ matrix considering that the high difference in degeneracy between Coⁿ⁺/Co^{(n ±1)+} spin states has been pointed out as a possible origin of the large S values.

2. Experimental

The studied 3D systems crystallize a perovskite structure, which, in dependence of the size of interpolated cation adopts either *Pm3m*, *Pbnm* or *R3c* symmetry, the 2D perovskites adopt *I4/mmm* symmetry. The samples La_{1-x}A_xCoO₃ ($0 \le x \le 0.5$; A=Ca,Sr or Ba) and Y₁. _xCa_xCoO₃($0 \le x \le 0.20$), La_{2-x}Sr_xCoO₄($0.5 \le x \le 1.5$) were prepared by a solid-state or nitrate decomposition method, respectively. The samples were synthesized either in oxygen (Co⁴⁺ rich), air (\sim Co³⁺) or argon (Co2+) atmosphere in order to achieve the oxygen stoichiometry. The X-ray diffraction analyses at room temperature confirmed that all studied samples were single phase. The electrical resistivity, thermoelectric power and thermal conductivity were measured in vacuum by a standard four terminal method in close cycle cryostat (T<300 K) or in air in the tubular furnace (T>300K), using home made set-ups. The magnetization M was measured using a SQUID (T<400 K) or torque (T>300 K) magnetometers.

3. Results and Discussion

3.1 Y1-xCaxCoO3 (0≤x≤0.20) system

Magnetic, structural and transport properties were investigated over a broad temperature region up to 1000 K. The temperature dependence of magnetic susceptibility is in Fig. 2 compared with lines marking the characteristic spin states. The magnetic susceptibility data shows that the Ca doping does not induce the significant change of magnetic behaviour and both Y-based samples, x=0.0 and x=0.1. preserve the low-spin ground-state up to ~400 K. Contrary to published data on LaCoO₃, the high temperature metal insulator transition is evidently not associated with a IS=>HS spin state transition but is accompanied only by the gradual increase to the intermediate spin state with S=1. Nonetheless, in distinction to the magnetic data the electrical resistivity and thermoelectric power are substantially decreased upon the Ca doping.



Fig. 3. The temperature dependence of the thermoelectric power (upper panel) and electrical resistivity (lower panel) of $Y_{1,x}Ca_xCoO_3$.

We suppose that, contrary to LS Na_xCoO_2 cobaltites, where the possible impact of spin and orbital degeneracy on the thermoelectric power was emphasized, the high value of thermoelectric power up to 700 K is dominantly influenced by a low concentration of charge carries.

3.2 La_{1-x}Sr_xCoO₃ system

We show in Fig. 4 the temperature dependence of the magnetic susceptibility for small Sr doping up to 0.1. We see that the incorporation of even very small concentration of formally Co^{4+} species via A^{2+} substitution leads to the appearance of FM interactions. and the ferromagnetism, competing with antiferromagnetic interactions in LaCoO₃, leads to spin glass behaviour.



Fig4. The temperature dependence of the DC magnetic susceptibility for $La_{1-x}Sr_xCoO_{3+\delta}$.

The magnetic fluctuations depress simultaneously the thermal conductivity, as evidenced in the Fig 5. Nonetheless the strongly depressed thermal conductivity below 300 K is observed also for pure LaCoO₃. The low temperature phononic peak of λ_{ph} is, however, completely wiped out for x>0.02.



Fig5. The temperature dependence of the thermal conductivity for $La_{1-x}Sr_xCoO_{3+\delta}$.

The origin of this behaviour resides as in the Jahn-Teller active IS Co^{3+} ($t_{2g}5e_g1$) dynamic modes (intermediate temperatures) as in a strong scattering due to fast hopping electrons between IS Co^{4+} ($t_{2g}4e_g1$) or LS Co^{4+} ($t_{2g}5e_g0$) and IS Co^{3+} ($t_{2g}5e_g1$) species (low temperatures). The association of the strongly depressed thermal conductivity with IS Co^{3+} is supported by comparison with the thermal conductivity for the LS $Y_{1-x}\text{Ca}_x\text{CoO}_3$ presented in Fig 6. Despite of a high disorder due to Ca for Y substitution (both size and mass mismatch) and high porosity of the sample the broad low temperature λ_{ph} peak is still present for x=0.1 sample.



Fig6. The temperature dependence of the thermal conductivity for $Y_{1-x}Ca_xCoO_3$.



Fig7. The temperature dependence of the thermoelectric power for $La_{1-x}Sr_xCoO_{3+\delta}$.

Finally we demonstrate in the Fig. 7 the temperature dependence of the thermoelectric power S for $La_{1-x}Sr_xCoO_{3+\delta}$. For the lowest Sr concentrations the electrical resistivity exceeds the minimum metallic conductivity (the Ioffe-Regel "metallic" limit ρ ~1-2 m Ω cm)

and is strongly temperature dependent pointing to the hopping mechanism of the charge carrier transport. In coherence the thermoelectric power is temperature weakly dependent and its the absolute value at room temperature corresponds to that deduced from a simple Heiks configurational entropy approximation;

$$S_{\text{Heiks}} = \frac{k_B}{e} \ln\left[\frac{1-x}{x}\right] \qquad [1]$$

Applying this formula e.g for x=0.005 we observe $S_{\text{Heiks}} = 455 \ \mu\text{VK}^{-1}$ which compares with $S_{exp} \sim 500 \ \mu\text{VK}^{-1}$, similarly for x=0.05 $S_{exp} \sim 250 \ \mu\text{VK}^{-1}$ matches with $S_{\text{Heiks}} \sim 257 \ \mu\text{VK}^{-1}$.

For the higher x the thermopower is reduced below the Heiks approximation and the behaviour approaches to that typical for the metals with a high carrier concentration. This corroborates with a strongly enhanced electrical conductivity approaching at 300 K to the Ioffe-Regel "metallic" limit.

3.3 N-type doping

We have attempted to introduce the n-type carriers in both LS (DyCoO₃) and IS (LaCoO₃) parent systems substitution of Co³⁺ by Ti⁴⁺ (3d⁰) was found successful with respect to sample purity. As evidenced in Fig. 8 for DyCo_{1-x}Ti_xO₃ the thermopower is negative below ~ 400 K confirming thus the formation of n-type charge carriers at low temperatures.



Fig8. The temperature dependence of the thermoelectric power for the Ti doped LS $DyCo_{1,x}Ti_xO_3$ up to 900 K (x=0.02,0.05). In the insets the thermal evolution of the electrical resistivity and thermal conductivity is shown.

The creation of n-type carriers conductivity is not, however, accompanied by commonly observed decrease of electrical resistivity – see the inset. This indicates that the N-type charge carriers posses a very low mobility which corroborates with their limited impact on phonon scattering –see the inset. Above 400 K the thermopower switches to positive values indicating the dominance of P-type carriers and when the temperature is further increased above the ~700 K the broad metalinsulator transition is observed and accompanied by a significant decrease of thermopower below ~100 μ VK⁻¹. As a possible explanation of temperature limited N-type conductivity we propose the electron hopping between the Co²⁺ (stabilized eventually in LS state via LS Co³⁺ and/or JT distorted IS Co³⁺ environment) and nonmagnetic LS Co³⁺. On the other hand the hopping between magnetic HS Co²⁺ and IS or HS Co³⁺ is strongly limited due to dominating antiferromagnetic interactions between these species. This picture is further supported on the study of 2D perovskites La_{2-x}Sr_xCoO₄ as shown in following section.

3.4 2D perovskites La2-xSrxCoO3 system

In order to check the possibility of both N- and Ptype conduction in a wide range using single prototype system we have performed the investigation of $La_{2-x}Sr_xCoO_4$ layered perovskites where, in principle, the Co formal valence can be varied from Co^{2.5+} up to Co^{3.5+}. The magnetic interactions are revealed in Fig.9 where the magnetic susceptibility of Co⁴⁺ rich side is presented. With the increasing Co4+ concentration, similarly as in case of 3D analogue, the Weiss constant increases and culminates at Θ ~170 K for x=1.4-1.5 indicating, together with the high value of the susceptibility and spontaneous moment, the long range ferromagnetic order. Simultaneously with increasing Θ , the spin only number S increases from minimum for x=1.0 (S-0.9) to S-1.2 for x=1.5. The Co^2 rich side is characterized by negative Weiss constant decreasing down to $\Theta \sim -200$ K for x=0.5 (Co²⁺ /Co³⁺=1) while the spin number increases up to S~1.8.



Fig9. The temperature dependence of the magnetic susceptibility for $La_{2-x}Sr_xCoO_4$ (1<x<1.5). In the inset the composition dependence of the spin number S and Weiss constant Θ is shown (data for 0.4<x<1.0 taken from Ref 6).

The increasing strength of antiferromagnetic interactions with decreasing cobalt valency is understandable presuming that the HS Co^{2+} species interact antiferromagnetically with surrounding IS or HS Co^{3+} network. The Co^{2+}/Co^{3+} system is electrically insulating and contrary to 3D cobalt perovskites, even all ferromagnetic samples remain insulating down to low temperatures.

The interesting insight in the carries subsystem offers the complex Figure 9 where the temperature dependence of the thermopower is presented. The most important feature is that for any studied composition and for any temperature region the negative thermoelectric power is not achieved. At first glance this is very surprising, as intuitively one should presume, as a consequence of electron addition, the N-type conduction when lowering the valence below Co³⁺. We speculate that this behaviour likely originates from the "immobility" of additional electrons associated with HS Co2+ species. On the other hand the introduction of Co4+ leads to substantial of electrical resistivity decrease accompanied simultaneously by decreased thermoelectric power, e.g. from ~175 μ VK⁻¹ (x=1.1) down to ~25 μ VK⁻¹ (x=1.5) at 300 K.



Fig10. The temperature dependence of the thermoelectric power for $La_{2x}Sr_xCoO_4$ (1<x<1.5) up to 900 K.

3.5 Co³⁺:Co⁴⁺=1:1

The role of mismatch was finally studied on 3D systems where the cobalt valence was fixed as $Co^{4+}/Co^{3+}=1/1$, i.e. where, supposing the hopping conduction, the simple Heiks configurational contribution is missing. In the Fig 11 we show the temperature dependence of thermopower for three cobaltites which differ significantly with respect to structural matching of La(Sr.Ca,Y)-O and Co-O bonds - starting from relaxed La0.5Sr0.5CoO3 (t=0.975), via La0.5CoO3 (t=0.95) and ending by highly distorted La_{0.3}Y_{0.2}Ca_{0.5}CoO₃ (t=0.94). The most conducting cubic La-Sr system posses at room temperature resistivity of ~0.3 mOcm and is ferromagnetic below ~ 250 K, for the highly distorted La-Y-Ca system the resistivity increases up to ~4.5 mΩcm and T_e is lowered below 100 K. At 900 K the difference in electrical resistivity is decreased to ~0.7 mOcm (La-Sr) and ~2.7 mOcm (La-Y-Ca), respectively. In order to explain the temperature weakly dependent and relatively high value of the thermopower for La-Y-Ca system we employ the generalised Heikes formula



Fig. 11. The temperature dependence of the thermoelectric power for $La_{0.5}Sr_{0.5}CoO_{3}$, $La_{0.5}Ca_{0.5}CoO_{3}$ and $La_{0.3}Y_{0.2}Ca_{0.5}CoO_{3}$ up to 900 K.

$$S = -\frac{k_B}{e} \ln \left[\beta_S \beta_O \frac{x}{1-x}\right] \quad [2]$$

supposing that dominating contribution to the thermopower is the configurational entropy of quasi-degenerate charge carriers. The additional $\beta_{s}\beta_{0}$ terms reflect the spin and orbital degeneracy (see Table 1) and $x = Co^{4+}/Co$, the fraction of Co⁴⁺ holes sitting on the Co sites, respectively.

This allows express $\beta_{S}\beta_{0}$ as the ratio $\frac{g_{3}}{g_{4}}$, where g_{3} and g_{4}

stand for the degeneracy of configurations for trivalent and tetravalent cobalt in the octahedral coordination, respectively.

Using this picture the hopping conductivity of the Co4+ rich perovskites can be interpreted supposing the IS spin state of both Co³⁺ and Co⁴⁺. In that case the holes of the t_{2g} character are involved and for identical Co³⁺ and Co4+ concentration, independently on the eventual Jahn-Teller distortion, the ratio between g_3 and g_4 = leads to $\beta_8\beta_0$ =3/4 in the formula [2]. This leads to $S \sim 24 \,\mu\text{VK}^{-1}$ what is indeed very close to the value experimentally observed in highly mismatched 3D and 2D cobaltites, containing 50 %Co⁴⁺, at high temperatures when hopping conductivity dominates - see Fig.10, 11. Nonetheless, when the structural distortion in 3D perovskite is decreased, the Co-O-Co bond is straightened and diffusive conduction of the itinerant or quasi-itinerant charge carriers eg electrons start to dominate ($\rho <<1$ m Ω cm). This situation corroborates with the ferromagnetic coupling, observed for the high enough Co4+ concentration favouring the kinetic energy gain of the eg electron linking the neighbouring Co ions. This makes the states before and after the electron transfer degenerate and corresponds to the $t_{2g}^5 \sigma^{*0.5}$ electronic configuration (by other words to 50% IS Co³⁺ and 50% LS Co⁴⁺). This leads to similar magnetic exchange mechanism as known for ferromagnetic manganites- so called DoubleExchange. In the ferromagnetic state the corresponding thermoelectric power should be negative due to the electron diffusion in a quarter filled band. Indeed, as shown in Fig 11 for ferromagnetic metallic La-Sr and La-Ca systems, this situation is experimentally observed.

4. Conclusions

The physical properties of various cobalt 3D and 2D perovskite cobaltites have been measured and analyzed. We summarize the complex experimental observation:

(i) independently on the cobalt valency the low spin ground-state of Co ions in $Ln_{1-x}A_xCOO_3$ is favoured by a small size of the interpolated cation. The LS state can be extended well above 300 K, promoting consequently the absolute value of thermoelectric power, (ii) the large depression of the thermal conductivity in $LnCoO_3$ is directly connected with the gradual stabilization of IS Co^{3+} species, the depressed thermal conductivity in $LnCoO_3$ is directly connected with the gradual stabilization of IS Co^{3+} species, the depressed thermal conductivity in $Ln_{1-x}A_xCoO_3$ is due to fast electron fluctuations between $LS(IS)Co^{4+}$ and IS Co^{3+} species, (iii) the absence of negative thermoelectric power in 2D $La_{2-x}Sr_xCoO_4$ and in 3D perovskites (at high temperatures only) is likely linked with the "immobility" of electrons associated with antiferromagnmetically coupled HS Co^{2+} species surrounded by magnetic Co^{3+} species, (iv) the generalised

Heikes formula is applicable in interpretation of the hopping conductivity in perovskite cobaltites.

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